

Table B.2 (continued)

AA 5059 Alustar	Sheet, strip, plate $3 < t < 50$	Al,Mg 5-6	0/H111	330	300	160	160	0,7	112	65
AA 5059 Alustar	Sheet, strip, plate $3 < t < 50$	Al,Mg 5-6	H32	370	300	270	160	0,7	112	65
EN AW-5383	Sheet, strip, plate $3 < t < 50$	Al,Mg 4,5 Mn 0,9	0/H111	290	290	145	145	0,7	102	59
EN AW-5383	Sheet, strip, plate $3 < t < 50$	Al,Mg 4,5 Mn 0,9	H32	305	290	220	145	0,7	102	59
Aluminium alloys (heat treatable) $\sigma_d = 0,7 \sigma_{yw}$ and $\tau_d = 0,58 \sigma_d$										
EN AW-6060	Profiles, bars, Tubes $3 < t < 25$	Al,Mg Si	T5,T6	190	95	150	65	0,7	46	26
EN AW-6061	Profiles, bars, Tubes $3 < t < 25$	Al,Mg1, Si Cu	T5,T6	260	165	240	115	0,7	81	47
EN AW-6061	Closed profiles	Al,Mg1, Si Cu	T5,T6	245	165	205	115	0,7	81	47
EN AW-6063	Profiles, bars, Tubes $3 < t < 25$	Al,Mg 0,7 Si	T5	150	100	110	65	0,7	46	26
EN AW-6063	Profiles, bars, Tubes $3 < t < 52$	Al,Mg 0,7 Si	T6	205	100	170	65	0,7	46	26
EN AW-6005A	Profiles, bars, Tubes $3 < t < 51$	Al,Si,Mg (A)	T5,T6	260	165	215	115	0,7	81	47
EN AW-6005A	Closed profiles $3 < t < 50$	Al,Si,Mg (A)	T5,T6	250	165	215	115	0,7	81	47
EN AW-6082	Profiles, bars, Tubes $3 < t < 25$	Al,Si 1,Mg,Mn	T5,T6	310	170	260	115	0,7	81	47
EN AW-6082	Closed profiles	Al,Si 1,Mg,Mn	T5,T6	290	170	240	115	0,7	81	47
EN AW-6106	Profiles, bars, Tubes $3 < t < 25$	Al,Mg,Si,Mn	T6	240	240	195	195	0,7	81	47
<p><sup>a</sup> The ultimate values are given for information only as the design stress is based on yield strength in welded conditions.</p> <p><sup>b</sup> The value of design stress is for welded aluminium. For unwelded aluminium (riveted or glued), <math>\sigma_d = \min (0,6\sigma_{uw} \text{ or } 0,9\sigma_{yw})</math> unwelded a.</p> <p>NOTE <math>\sigma_u</math> and <math>\sigma_y</math> are tensile stresses.</p> <p>The value of <math>E</math> modulus of metal is required in some formulas (e.g. Table A.9, A.11 &amp; A.12, etc.) and, unless specifically documented, the default following values may be used:</p> <p>Mild steel <math>E = 210\,000 \text{ N/mm}^2</math> Aluminium alloys <math>E = 70\,000 \text{ N/mm}^2</math>.</p>										

## Annex C (normative)

### FRP laminates properties and calculations

#### C.1 Status of this Annex

This Annex shall be used for the analysis methods 1 to 3 of [Table 2](#). For the analysis method 5 (FEM), other documented values may be used but it should be checked that their values do not differ from the ones in this Annex by a large margin.

#### C.2 Determination of the mechanical properties

##### C.2.1 Tests and test standards

Mechanical properties to be used as input in determining the bending moment, stiffness and shear capabilities of FRP laminates and stiffeners may be derived either by testing of representative samples using the appropriate ISO or ASTM test standards or by calculation or by a combination of the two.

**Table C.1 — Examples of test standard references and specific tests**

1-Examples of test standards
Tensile properties: ISO 527-4, ISO 527-5 Flexural properties: ISO 178 Compressive properties: ISO 14126 Inplane shear properties: ISO 14129 Interlaminar shear stress: ISO 14130 Through-thickness « flatwise » tensile properties: ASTM D7291
2-General application of the above standards
Where an International Standard does not exist, a national standard may be used instead. The number of samples to be tested shall be as laid down in international or national standards but shall not be less than five samples for any given property.  When determining the flexural strength, the gel coat side of the specimen shall be stressed in tension. Unless specifically stated in the test standard, the mechanical properties used in the calculations shall be corrected from test values as follows: <ul style="list-style-type: none"> <li>— for strength 90 % of the mean ultimate strength or the mean value minus two standard deviations whichever is the lesser;</li> <li>— for elastic modulus, the mean value.</li> </ul>

Table C.1 (continued)

3-Alternative method for testing compression strength of UD
<p>It is often difficult to apply ISO 14126, particularly for carbon-based UD</p> <p>An alternative method based on four-point bending tests can be used to measure compressive stress of unidirectional composites (glass, carbon fibres).</p> <p>Comparing to « pure » compressive tests (ISO 14126), the advantages of four-point bending tests are the followings:</p> <ul style="list-style-type: none"> <li>— sample geometry (no end tabs, tolerances);</li> <li>— common tooling's;</li> <li>— adapted for thick laminates;</li> <li>— possibility to measure ultimate compressive stress of unidirectional layers inside real scantlings.</li> </ul> <p>In order to prevent early damages under load points, bi-axial layers (0/90 or +-45) on sample facings and load tabs can be used.</p> <p>Sample failures are validated when the failure occur at the upper face between the 2 load points.</p> <p>From ply thickness measurements, an ultimate compressive strength in the UD is then calculated using the recorded force at failure.</p> <p>These tests may be conducted using the specifications of ASTM D6272 standard.</p>

As a minimum, the fibre mass fraction ( $\psi$ ) shall be measured by weighing a resin-consolidated panel of known fibre mass, see Example 1. Resin ignition tests may also be used. The panels used for this purpose shall be representative of the as built quality. Where it is not practical to take suitably sized panels from the actual or previous craft, special care must be taken to ensure that laboratory made samples are representative.

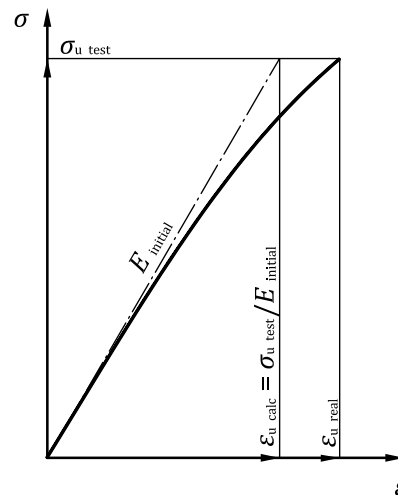
[Table C.2](#) gives the relations between the mass fraction  $\psi$ , the volume fraction  $\phi$ , the ratio  $t/w$  and laminate density  $\rho$ .

NOTE 1 The above requirements aim at taking the appropriate steps to ensure that the mechanical capability (not simply mechanical properties, but also taking geometry into account) and properties of the as-built laminate are equal or superior to those at the design stage.

### C.2.2 Topics on tests and calculation

Many CLT software consider a linear behaviour  $\sigma = E \times \varepsilon$ , which is not true in reality: when testing a laminate, one can see that, after an initial linear behaviour, the stress/strain plot gets curved. This is due to micro cracks or damages occurring after the failure of some, non-critical, plies.

It is practically difficult to measure  $\varepsilon_u$  real, and when the initial elastic modulus,  $E_{\text{initial}}$ , and  $\sigma_{u \text{ test}}$  are measured, one shall take  $E_{\text{calculation}} = E_{\text{initial}}$  and  $\varepsilon_{\text{calc}} = \frac{\sigma_{\text{test}}}{E_{\text{initial}}}$ , see [Figure C.1](#).

Figure C.1 — Determination of  $\varepsilon_u$ Table C.2 — Values of  $t/w$ , composite density  $\rho_c$  according to fibre content by volume  $\phi$  or mass  $\psi$ 

$\psi = \frac{\phi \times \rho_f}{\phi \times \rho_f + (1 - \phi) \times \rho_m}$		$\phi = \frac{\psi}{\psi + (1 - \psi) \times \frac{\rho_f}{\rho_m}}$	$\frac{t}{w} = \frac{1}{\phi \times \rho_f}$
$\rho_c = \rho_f \times \phi + \rho_m \times (1 - \phi) \text{ or } \rho_c = \frac{\rho_f \times \rho_m}{\rho_f + \psi (\rho_m - \rho_f)}$			$t = \frac{w}{\rho_f \times \rho_m} \left( \frac{\rho_f}{\psi} + \rho_m - \rho_f \right)$
$t$	Thickness of the laminate		mm
$w$	Dry mass of fibre		kg/m <sup>2</sup>
$\phi$	Fibre content by volume in the laminate (dry fibre volume/laminate volume) <sup>a</sup>		1
$\psi$	Fibre content by mass in the laminate (dry fibre mass/laminate mass).		1
$\rho_f, \rho_m$	Density, respectively of fibre and matrix, may be taken from <a href="#">Table C.2</a> or from manufacturer's information		kg/m <sup>3</sup>
<sup>a</sup> For guidance purposes see also <a href="#">Table C.7</a> .			

### C.2.3 Use of flexural strain and strength

The values of flexural stresses as tested with ISO 178 or equivalent give significantly high stresses compared with tensile/compression. Reference [25] quotes  $\sigma_{uf} = k_G \times \sigma_{ut} / (1 + \sigma_{ut}/\sigma_{uc})$  for GRP with  $k_G$  ranging from 2,5 to 3 according to the type of laminates, similar values for Carbon and Aramid. This high values of  $\sigma_{uf}$  induced that ISO 12215-5:2008 required single skin laminates using  $\sigma_f$  that were always thinner than the ones required by the ply by ply analysis of [Annex H](#), using  $\sigma_t$  or  $\sigma_c$ . However as the -thinner- laminates designed with  $\sigma_f$  did not prove to be underbuilt in practice, this revision document allows the use of  $\pm\sigma_f$  instead of  $\sigma_t/\sigma_c$  for the ply by ply analysis of single skin laminates in [Annex H](#). The reason is that these laminates are likely to operate in the large deflection regime, particularly under high slam pressures. As the scantling formulas are based on the more conservative small deflection theory, the use of in plane ultimate strains may be a case of "double penalty". Sandwich panels, being stiffer are more likely to be operating within the small deflection regime and hence the use of inplane strains is appropriate. CLT theory method usually only apply  $\sigma_t/\sigma_c$ . [Tables C.6](#) and [C.9/C.10](#) detail the values of  $\varepsilon_{uf}$ .

### C.2.4 Mechanical properties for the simplified method

The "simplified" method described in [11.2](#) and [Table A.5](#) only considers Glass Reinforced Plastics (not Carbon composites). Unless derived from tests, according to [C.2](#), the mechanical properties shall be

derived from [Tables C.6 to C.10](#). These tables do not give the values for one "thick" layer of a mixing of the plies given in [Table C.6](#), like, for example, Mat/Roving. [Clause C.4](#) and [Table C.11](#) give some examples on such calculations.

### C.2.5 Elastic constants using 'CLT' method (classical laminate theory)

In the 'CLT' method, the elastic constants are derived from manufacturers' fibre and resin properties for a known fibre volume content using well-established and empirically modified rule of mixtures formulas. It is not necessary to provide these formulas within the document since they are documented in the "Law of mixtures" and Classical Lamination theory (CLT). As for single skin, the flexural stress is used, it is recommended to replace the design strains  $\varepsilon_{ut}$  or  $\varepsilon_{uc}$  by  $\pm\varepsilon_{uf}$  otherwise the scantlings could be conservative compared to the simplified or enhanced method.

**NOTE** For guidance, the Bureau Veritas or DNV-GL publications listed in the bibliography were among the source references used in developing [Tables C.5](#) and [C.6](#). Elastic constants of woven roving, double-bias (+45/-45), etc. may be obtained separately from the [A] submatrix by combining unidirectional plies at the appropriate orientation (a symmetrical layup being recommended to eliminate in-plane/out of plane coupling).

### C.2.6 Elastic constants using 'SRM' method (simplified regression method)

For the builders or designers not wishing to use the CLT method, the simpler Simplified Regression Method (SRM) is proposed, both methods work similarly for cross plies, but treat angle plies differently, see details in [C.2.5](#). [Table C.4](#) shows physical and mechanical properties used in the CLT method, and [Table C.5](#) gives the main mechanical properties of UD and of several "typical" plies/multiplies.

[Table C.3](#) sums up the "flow chart" of the procedure to follow in either SRM/Ply stack analysis or CLT.

**Table C.3 — Procedure to obtain mechanical properties and allowable bending moment and shear forces for single skin or sandwich plating**

STAGE 1-PLY ELASTIC CONSTANTS AND BREAKING STRAINS	
Obtained from; Test data or Simplified regression formulas ( <a href="#">Tables C.4</a> to <a href="#">C.8</a> tabulated in <a href="#">Tables C.9</a> & <a href="#">C.10</a> ) or Other verified formulae ( <a href="#">Tables C.2</a> & <a href="#">C.5</a> )	
STAGE 2-MODIFY PROPERTIES TO REFLECT BOAT BUILDING CHARACTERISTICS $k_{BB}$ ( <a href="#">Table 16</a> )	
Obtained from; Same formulas as above but modified following tests and details and by $k_{BB}$ in <a href="#">Table 16</a>	
STAGE 3-OBTAIN LAST PLY FAILURE BENDING MOMENT AND SHEAR FORCES	
For single skin laminates use $\sigma_d = 0,5 \pm \sigma_{uf} \times k_{BB}$ For sandwich laminates, depending on stress sense use $\sigma_d = \sigma_{ut}$ or $\sigma_{uc}$ or wrinkling $\times k_{BB}$ Obtain flexural stiffness for deflection checks	
OPTION 1	OPTION 2
<a href="#">Annex H</a> : Simplified laminate stack method	CLT (Classical Lamination Theory) <sup>a</sup>
Multiply the previous design stress by $k_{AM} = 0,95$ for FRP according to <a href="#">Table 16</a> (Enhanced method) Caution: the strains and stresses are along fibre	Multiply the previous design stress by $k_{AM} = 1$ for FRP according to <a href="#">Table 16</a> (developed method) Caution; CLT software may not always check wrinkling, then a manual check is needed For DBx and Qx input as angled WR combination and transformed into 1-2 system for assessment Failures are based on any of the generally recognized failure envelope formulas.

Table C.3 (continued)

STAGE 4-OBTAIN ACTUAL vs REQUIRED LAST PLY FAILURE BENDING MOMENT AND SHEAR FORCES	
Use <a href="#">Annex H</a> or CLT software to find last ply failure stress (see NOTE), calculate design stress per <a href="#">Table 17</a> and find the compliance factor $CF = \sigma_{\text{design}} / \sigma_{\text{actual}}$ that shall be $\geq 1$	
<sup>a</sup> For sandwich check whether CLT software can obtain shear force internally (need to have transverse shear stiffness with $4 \times 4$ matrix) otherwise use method in <a href="#">Annex H</a> for sandwich.	
NOTE The last ply failure is usually the first ply failure, but this is not always the case, particularly when one mix stiff (carbon) and non stiff plies (GRP), or when using UD plies at $90^\circ$ from mains stress direction. For example in a ply with 90 % UD in the sense of maximal stress and 10 % UD perpendicular to it, the transverse ply fails first but 90 % of the strength remains.	

## C.2.7 Final mechanical properties

This Annex proposes several methods to define the mechanical properties of the laminate, but their final values depend significantly on the “as built” quality of the material achieved by the boatbuilder. Therefore, the final “design” mechanical properties used for calculation of composites in [Table 17](#) needs to be adjusted by the factors  $k_{BB}$  and  $k_{AM}$  defined in [Table 16](#).

[Tables C.4](#) to [C.10](#) may be implemented for other fibres or matrices.

Table C.4 — Physical and mechanical properties of fibres and matrices

			Reinforcement fibres			Matrices
			E Glass	Aramid	Carbon HS	Polyester/epoxy
Specific gravity $\rho$ ( $\rho_f$ or $\rho_m$ )	t/m <sup>3</sup>		2,56	1,44	1,78	1,2
Elastic modulus $E$ ( $E_{f1}$ , $E_{f2}$ or $E_m$ )	$E_{f1} //$ Fibres	N/mm <sup>2</sup>	73 000	124 000	235 000	3 300
	$E_{f2} \perp$ Fibres	N/mm <sup>2</sup>	73 000	6 900	20 000	3 300
Shear modulus $G$ ( $G_f$ or $G_m$ )	N/mm <sup>2</sup>		30 000	2 800	50 000	1 222
Poisson's ratio $\nu$ ( $\nu_f$ or $\nu_m$ )	1		0,22	0,36	0,27	0,32

NOTE The standard formulas in the CLT method are either exact linear (major modulus and Poisson's ratio) or very nearly quadratic (minor modulus and shear modulus). The CLT method requires input data which may not be readily available from manufacturers, such as transverse modulus of fibre and resin Poisson's ratio. Furthermore, there is no fixed relationship between for example modulus of polyester versus that of epoxy which holds for all commercially available resin. Consequently, the source references above together with predictions from formulas used either using Reference [\[24\]](#) or by the National Physical Laboratory (UK) (which include empirical correction factors) have been used for generic fibres and resin as defined in [Table C.4](#).

**Table C.5 — Formulas for UD and laminates of [Tables C.6](#) and [C.7](#)**

1-Theoretical formulas for UD			
$E_{UD1} = 0,975 \times [E_{f1} \times \phi + E_m \times (1 - \phi)]$		$E_{UD2} = E_m \frac{1 + \zeta \times \eta_E \times \phi}{1 - \eta_E \times \phi}$ with $\zeta = 1$ and $\eta_E = \frac{E_{f2} / E_m - 1}{E_{f2} / E_m + \zeta}$	
$G_{UD12} = G_m \frac{1 + \zeta \times \eta_G \times \phi}{1 - \eta_G \times \phi}$ with $\zeta = 1$ and $\eta_G = \frac{G_f / G_m - 1}{G_f / G_m + \zeta}$		$\nu_{UD} = \nu_f \times \phi + \nu_m \times (1 - \phi)$ Poisson's ratio	
$\tau_{UIL} = 22,5 - \frac{33\phi}{\phi + 0,89}$ interlaminar shear stress		$\nu_{UD\ 21} = \nu_{UD\ 12} \frac{E_{UD2}}{E_{UD1}}$	
The above formulas are derived from Halpin-Tsai formulas, where $\phi$ is the fibre content in volume, and other variables defined in <a href="#">Table C.4</a>			
2-Formulas for other laminates: CSM; biaxial 0/90 (BD+); double bias $\pm 45$ (DB $\times$ ), and quadriaxial (Q $\times$ )			
Except for E glass chopped strand mat, the formulas apply to any "building" FRP fibre (Glass, carbon, Aramid, etc.)			
Material	Young's modulus	Shear modulus	Major Poisson's ratio
E Glass Chopped strand mat	$E_{CSM} = 3/8 E_{UD1} + 5/8 E_{UD2}$	$G_{CSM} = 1/8 E_{UD1} + 1/4 E_{UD2}$	$\nu_{CSM} = E_{CSM} / 2 G_{CSM} - 1$
BD+ 0/90 Biaxial or woven roving	$E_{BD+} = 0,5 (E_{UD1} + E_{UD2})$	$G_{BD+} = G_{UD12}$	$\nu_{BD+} = \nu_{UD12} \frac{E_{UD2}}{E_{BD+}}$
DB $\times$ $\pm 45^\circ$ Double bias	$E_{DBx} = \frac{4E_{BD+}}{\frac{E_{BD+}}{G_{BD+}} + 2(1 - \nu_{BD+})}$	$G_{DBx} = \frac{E_{BD+}}{2(1 + \nu_{BD+})}$	$\nu_{DBx} = \frac{E_{DBx}}{4E_{BD+}} \left[ \frac{E_{BD+}}{G_{BD+}} - 2(1 - \nu_{BD+}) \right]$
Q $\times$ Quadriaxial 0/45/90/-45°	$E_{QX} = 0,5[A_{11} - A_{12}^2/A_{11}]$	$G_{QX} = 0,5 (G_{BD+} + G_{DBx})$	$\nu_{QX} = A_{12}/A_{11}$
with $A_{11} = \frac{E_{BD+}}{1 - \nu_{BD+}^2} + \frac{E_{DBx}}{1 - \nu_{DBx}^2}$ and $A_{12} = \frac{\nu_{BD+} \times E_{BD+}}{1 - \nu_{BD+}^2} + \frac{\nu_{DBx} \times E_{DBx}}{1 - \nu_{DBx}^2}$			

**C.2.8 Breaking strains — Both methods CLT or SRM**

The breaking strains, to be used in [Table C.8](#) are given in [Table C.6](#).

Table C.6 — Breaking strains in %

Breaking strains <sup>a</sup> (ultimate strength/initial <i>E</i> modulus) in %			
Type of fibre & resin		E Glass & polyester	HS Carbon & epoxy
$\varepsilon_{ufi} = k_G \varepsilon_{uti} / (1 + \varepsilon_{uti} / \varepsilon_{uci})$ with $i = 1$ or $2$ and $k_G^b = 2,50$ or $2,94$ , see columns 3 and 4.		$k_G = 2,50$	$k_G = 2,94$
Unidirectional quoted "UD"	$\varepsilon_{ut1}$	1,90	1,00
	$\varepsilon_{ut2}$	0,50	0,50
	$\varepsilon_{uc1}$	1,40	0,70
	$\varepsilon_{uc2}$	1,40	1,90
	$\varepsilon_{uf1}^b$	2,02	1,21
	$\varepsilon_{uf2}^c$	0,92	1,16
	$\gamma_{u12}$	1,70	1,50
CSM Chopped strand mat	$\varepsilon_{ut}$	1,35	Not applicable
	$\varepsilon_{uc}$	1,70	
	$\varepsilon_{uf}^b$	1,88	
	$\gamma_{u12}$	2,00	
WR/bidirectional 0/90° quoted "BD+"	$\varepsilon_{ut}$	1,55	1,00
	$\varepsilon_{uc}$	1,40	0,70
	$\varepsilon_{uf}^b$	1,84	1,21
	$\gamma_u$	1,70	1,40
Double bias ±45 quoted "DB×°"	$\varepsilon_{ut}$	1,06	0,77
	$\varepsilon_{uc}$	1,02	0,75
	$\varepsilon_{uf}^b$	1,30	1,12
	$\gamma_u$	1,80	1,02
Quadriaxial 0/45/90/_45 quoted "Q×°"	$\varepsilon_{ut}$	1,30	0,92
	$\varepsilon_{ut}$	1,20	0,74
	$\varepsilon_{uf}^b$	1,56	1,21
	$\gamma_u$	1,70	1,02

<sup>a</sup> Design strain (%) =  $0,5 \times$  breaking strain. Design stress =  $0,5 \times$  Associated modulus  $\times$  breaking strain/100 Associated modulus means use  $E_{UD2}$  with  $\varepsilon_{UC2}$  to obtain compressive strength perpendicular to fibres for a unidirectional,  $G_{BX}$  with  $\gamma_U$  to obtain the shear strength for a biaxial, etc.

<sup>b</sup> The experimental factor  $k_G$  is proposed by Green in Reference [25] to correlate flexural strain with tensile and compressive strains in the fibre direction for UD and generally in composites.

<sup>c</sup> The value in b above has been applied to transverse strains on UD, but this is pending validation,  $\sigma_{uf2}$  being anyway  $\ll \sigma_{uf1}$ .

NOTE Table C.6 is based on published values in two classification rules, ISO 12215-5:2008 and data supplied by industry.

NOTE Tables C.6, C.7, C.9 and C.10 are only computed for *E* glass/polyester and *HS* Carbon/epoxy. Other building fibres (Other type of glass or carbon, Aramid, etc) or resins may be used provided documented values are used.

## C.2.9 Practical use of CLT & SRM methods

### C.2.9.1 Preliminary

When using the CLT method, panel coordinate system strains ( $\varepsilon_x$ ,  $\varepsilon_y$  and  $\gamma_{xy}$ ) are transformed into individual ply coordinate system strains ( $\varepsilon_1$ ,  $\varepsilon_2$  and  $\gamma_{12}$ ) and hence a double-bias cloth is transformed into a WR/BD+ and the above strains may be used.



In SRM method, there is no transformation and it is necessary to determine  $E_{DBx}$  etc. for use in the laminate stack (see [Annex H](#)), it is also necessary to determine the breaking strain at 45 degrees to the fibres.

For built-in panels, the traditional approach is to use a single uniaxial stress or strain load case. Using the maximum stress or strain criterion, a double bias subjected to such a strain fails when;  $\varepsilon_{DB} = 2 \gamma_{U12} [G_{BX}/E_{DB}]$  (with  $\gamma_{U12}$  taken from [Table C.7](#) for WR/BD+).

### C.2.9.2 Use of CLT and SRM methods

The Classical Lamination Theory (CLT) method is intended for the ones using complex layups, often including UD plies at 0 and 90 degrees and/or triaxial or unbalanced biaxial cloths (different fibre mass in warp and weft) as well as asymmetrical layups up which generate  $D_{16}$ ,  $D_{26}$  and  $B_{ij} \neq 0$  in the stiffness matrix, therefore introducing complex coupling effects.

Users may apply SRM method properties as input into validated CLT software (commercial or in-house) with the load vector determined using the simple methods of the document.

However, when using UD-90 plies which generally have low tensile strain ( $\varepsilon_{ut2}$ ) as indicated in [Table C.6](#), CLT or [Annex H](#) are liable to give a low bending capability value as the method underpinning this document is first ply to fail. Analysing the stack as last ply to fail is normally outside the scope of this Annex.

The SRM method is intended for users who generally use combinations of biaxial, double bias and quadriaxial with or without mat, to achieve a reasonably balanced laminate.

The simple formulae for bending moments and shear forces using this standard generally do not consider more than one significant stress acting at any given point, i.e. the load vector generally consists of just  $M_x$  or  $M_y$  or  $N_{xy}$ ). This permits a laminate to be analysed using the laminate stack method outlined in [Annex H](#).

### C.2.10 Ply thickness

Ply theoretical thickness shall be calculated using the formulas of [Table C.2](#) (top right cells), or according either to volume fraction  $\phi$ , or mass fraction  $\psi$  or using the pre-computed values of [Table C.7](#), [C.9](#) or [C.10](#)

[Table C.7](#) considers that fibre content in volume  $\phi$  is mainly connected to the lamination process, and the fibre content in mass  $\psi$  is then calculated from the density of fibre and matrix (resin).

**Table C.7 — "Guidance values" for fibre content**

Lamination process	Material	Fibre content in volume $\phi$	Fibre content in mass $\psi$ , $t/w$ and composite density $\rho_c$					
			Glass $\rho_f = 2,56$			Carbon HR $\rho_f = 1,78$		
			$\psi$	$t/w$	$\rho_c$	$\psi$	$t/w$	$\rho_c$
Hand layup simple surface	CSM	0,167	0,300	2,34	1,43	—	—	—
	Woven Roving	0,300	0,478	1,302	1,61	0,389	1,87	1,37
	Rovimat	0,246	0,410	1,588	1,53	—	—	—
	Multidirectional	0,319	0,500	1,225	1,63	0,410	1,76	1,39
	Unidirectional	0,364	0,550	1,073	1,70	0,459	1,54	1,41
Hand layup Complex surface	CSM	0,134	0,248	2,924	1,38	—	—	—
	Woven Roving	0,240	0,403	1,628	1,53	0,319	2,34	1,34
	Rovimat	0,197	0,343	1,985	1,47	0,267	2,85	1,31
	Multidirectional	0,255	0,422	1,531	1,55	0,337	2,20	1,35
	Unidirectional	0,291	0,467	1,341	1,60	0,378	1,93	1,37

These values are given as a guide only and are considered achievable by the industry, but it is the responsibility of the builder to check the values that his building methods are currently achieving. For complex surfaces, the fibre content in volume is 80 % of the ones for simple surfaces.

Table C.7 (continued)

Lamination process	Material	Fibre content in volume $\phi$	Fibre content in mass $\psi$ , $t/w$ and composite density $\rho_c$					
			Glass $\rho_f = 2,56$			Carbon HR $\rho_f = 1,78$		
			$\psi$	$t/w$	$\rho_c$	$\psi$	$t/w$	$\rho_c$
RTM ECO	Any material	0,135	0,250	2,894	1,38	0,188	4,16	1,28
Infusion	CSM	0,21 - 0,30	0,36 - 0,48	1,86 - 1,30	1,49 - 1,61	0,28 - 0,39	2,68 - 1,87	1,32 - 1,37
	Woven Roving	0,42 - 0,50	0,61 - 0,68	0,93 - 0,78	1,77 - 1,88	0,52 - 0,60	1,34 - 1,12	1,44 - 1,49
	UD/Multidirectional	0,45 - 0,53	0,64 - 0,71	0,64 - 0,71	1,81 - 1,92	0,54 - 0,63	1,25 - 1,06	1,46 - 1,61
Prepreg void	UD/Multidirectional	0,530	0,706	0,737	1,92	0,626	1,06	1,51
Prepreg autoclave	UD/Multidirectional	0,530	0,706	0,737	1,92	0,626	1,06	1,51

These values are given as a guide only and are considered achievable by the industry, but it is the responsibility of the builder to check the values that his building methods are currently achieving. For complex surfaces, the fibre content in volume is 80 % of the ones for simple surfaces.

### C.3 Final calculation of E, G and ultimate stress

#### C.3.1 General calculation

Table C.8 — Final calculation of E, G,  $\sigma_u$  or  $\tau_u$ \*

$E = E_{\text{TABLE C 5}}$	$\sigma_u = E_{\text{TABLE C 5}} \times \varepsilon_{\text{TABLE C 6}}$
$G = G_{\text{TABLE C 6}}$	$\tau_u = G_{\text{TABLE C 5}} \times \gamma_{\text{TABLE C 6}}$

The calculations of [Table C.8](#) are pre-computed in [Table C.9](#) for Glass composites and in [Table C.10](#) for High strength carbon composites.

**CAUTION** — The design stress defined in [Table 17](#) shall then be determined using, the relevant value of  $k_{AM}$  and  $k_{BB}$  from [Tables 15 & 16](#).

#### C.3.2 Builder's responsibility

The use of any property data given in this Annex does not imply that these are achievable in practice for any particular craft. It is entirely the responsibility of the builder or his representative to demonstrate this. The tables giving mechanical properties in this Annex shall be adjusted by the factor  $k_{BB}$  of [Table 16](#) and used with care. It is subjective in nature and compliance with the table does not imply any guarantee that mechanical properties taken from this Annex are achieved for any particular craft in any particular location.

Table C.9 — Computed values of Table C.5 to C.8 for glass laminates (with  $k_{BB} = 1$ )

E GLASS																		
$\rho_f = 2,56$ $\rho_m = 1,2$ Ef1= 73 000 Ef2= 73 000 Em= 3 300 Gf= 30 000 Gm= 1 222			UD $\nu_f = 0,22$ $\nu_m = 0,32$ $\xi = 1$ $\eta E = 0,913$ $\eta G = 0,922$				CSM (Mat) $G_{CSM} = 1/8 E_{UD1} + 1/4 E_{UD2}$ $\nu_{CSM} = E_{CSM}/G_{CSM} - 1$			BD+ 0/90°			$\pm 45^\circ$ DBx Double bias			Quadraxial 0/45/90/-45		
$\phi$ Vol	$\psi$ Mass	$t/w$ 1/(( $\phi^2 \rho_f$ ))	$E_{UD1}$	$E_{UD2}$	$G_{UD12}$	$\nu_{UD12}$	$E_{CSM}$	$G_{CSM}$	$\nu_{CSM}$	$E_{BD+}$	$G_{BD+}$	$\nu_{BD+}$	$E_{DBx}$	$G_{DBx}$	$\nu_{DBx}$	$E_{Quad}$	$G_{Quad}$	$\nu_{Quad}$
0,140	0,258	2,79	12 732	4 268	1 584	0,31	7 442	2 658	0,40	8 500	1 584	0,15	4 817	3 684	0,52	6 911	2 634	0,31
0,160	0,289	2,44	14 091	4 430	1 645	0,30	8 053	2 869	0,40	9 260	1 645	0,15	5 047	4 042	0,53	7 459	2 844	0,31
0,167	0,300	2,34	14 566	4 488	1 667	0,30	8 267	2 943	0,40	9 527	1 667	0,14	5 129	4 168	0,54	7 653	2 917	0,31
0,180	0,319	2,17	15 450	4 599	1 708	0,30	8 668	3 081	0,41	10 024	1 708	0,14	5 282	4 402	0,55	8 013	3 055	0,31
0,200	0,348	1,95	16 809	4 775	1 774	0,30	9 288	3 295	0,41	10 792	1 774	0,13	5 523	4 764	0,56	8 572	3 269	0,31
0,220	0,376	1,78	18 168	4 960	1 844	0,30	9 913	3 511	0,41	11 564	1 844	0,13	5 770	5 127	0,56	9 137	3 485	0,31
0,240	0,403	1,63	19 527	5 153	1 916	0,30	10 544	3 729	0,41	12 340	1 916	0,12	6 025	5 491	0,57	9 708	3 704	0,31
0,260	0,428	1,50	20 886	5 356	1 992	0,29	11 180	3 950	0,42	13 121	1 992	0,12	6 289	5 858	0,58	10 285	3 925	0,31
0,280	0,453	1,40	22 246	5 568	2 072	0,29	11 822	4 173	0,42	13 907	2 072	0,12	6 562	6 226	0,58	10 869	4 149	0,31
0,300	0,478	1,30	23 605	5 792	2 156	0,29	12 471	4 398	0,42	14 698	2 156	0,11	6 846	6 595	0,59	11 460	4 376	0,31
0,319	0,500	1,22	24 896	6 014	2 240	0,29	13 095	4 616	0,42	15 455	2 240	0,11	7 126	6 949	0,59	12 028	4 594	0,31
0,340	0,524	1,15	26 323	6 273	2 337	0,29	13 792	4 859	0,42	16 298	2 337	0,11	7 449	7 341	0,59	12 664	4 839	0,31
0,360	0,545	1,09	27 682	6 534	2 436	0,28	14 465	5 094	0,42	17 108	2 436	0,11	7 770	7 717	0,60	13 280	5 076	0,31
0,380	0,567	1,03	29 041	6 809	2 539	0,28	15 146	5 332	0,42	17 925	2 539	0,11	8 107	8 095	0,60	13 904	5 317	0,31
0,400	0,587	0,98	30 401	7 100	2 649	0,28	15 838	5 575	0,42	18 750	2 649	0,11	8 460	8 476	0,60	14 539	5 563	0,31
0,420	0,607	0,93	31 760	7 409	2 766	0,28	16 540	5 822	0,42	19 584	2 766	0,11	8 831	8 860	0,60	15 185	5 813	0,31
0,440	0,626	0,89	33 119	7 736	2 889	0,28	17 254	6 074	0,42	20 427	2 889	0,10	9 222	9 247	0,60	15 844	6 068	0,31
0,460	0,645	0,85	34 478	8 083	3 021	0,27	17 981	6 331	0,42	21 281	3 021	0,10	9 634	9 637	0,59	16 515	6 329	0,30
0,480	0,663	0,81	35 837	8 454	3 161	0,27	18 723	6 593	0,42	22 145	3 161	0,10	10 069	10 031	0,59	17 202	6 596	0,30
0,500	0,681	0,78	37 196	8 849	3 311	0,27	19 479	6 862	0,42	23 023	3 311	0,10	10 530	10 429	0,59	17 904	6 870	0,30
0,530	0,706	0,74	39 235	9 495	3 556	0,27	20 647	7 278	0,42	24 365	3 556	0,10	11 276	11 034	0,59	18 992	7 295	0,30

E GLASS																									all plies inter laminar stress
$\rho_m = 1,2$ $\rho_f = 2,56$ Ef1= 73 000 Ef2= 73 000 Em= 3 300 Gf= 30 000 Gm= 1 222	UD						CSM				BD+ 0/90°				$\pm 45^\circ$ DBx Double bias				Quadrax 0/45/90/-45						
$k_G = 2,50$ $\epsilon_{ufi}=k_G \epsilon_{uti}/(1+\epsilon_{uti}/\epsilon_{uci})$ with i=1 or 2							$\epsilon_{uf}=k_G \epsilon_{ut}/(1+\epsilon_{ut}/\epsilon_{uc})$				$\epsilon_{uf}=k_G \epsilon_{ut}/(1+\epsilon_{ut}/\epsilon_{uc})$				$\epsilon_{uf}=k_G \epsilon_{ut}/(1+\epsilon_{ut}/\epsilon_{uc})$				$\epsilon_{uf}=k_G \epsilon_{ut}/(1+\epsilon_{ut}/\epsilon_{uc})$						
----- Values of strains (%) -----																									
$\epsilon_{U11}$ $\epsilon_{U12}$ $\epsilon_{Uc1}$ $\epsilon_{Uc2}$ $\epsilon_{Uf1}$ $\epsilon_{Uf2}$ $\gamma_{U12}$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	$\epsilon_{U1}$ $\epsilon_{Uc}$ $\epsilon_{Uf}$ $\gamma_U$	
1,90 0,50 1,40 1,40 2,02 0,92 1,70	1,35 1,70 1,88 2,00	1,55 1,40 1,84 1,70	1,06 1,02 1,30 1,80	1,30 1,20 1,56 1,70																					
----- Values of stresses (N/mm <sup>2</sup> ) -----																									
$\phi$ $\psi$ t/w	$\sigma_{U11}$ $\sigma_{U12}$ $\sigma_{Uc1}$ $\sigma_{Uc2}$ $\sigma_{Uf1}$ $\sigma_{Uf2}$ $\tau_U$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\sigma_{U1}$ $\sigma_{Uc}$ $\sigma_{Uf}$ $\tau_{UST}$	$\tau_{il}$	
Vol Mass 1/( $\phi \cdot \rho_f$ )																									
0,140 0,258 2,79	242 21 178 60 257 39 27	100 127 140 53	132 119 156 27	51 49 63 66	90 83 108 45	18,0																			
0,160 0,289 2,44	268 22 197 62 284 41 28	109 137 151 57	144 130 170 28	53 51 66 73	97 90 116 48	17,5																			
0,167 0,300 2,34	277 22 204 63 294 41 28	112 141 156 59	148 133 175 28	54 52 67 75	99 92 119 50	17,3																			
0,180 0,319 2,17	294 23 216 64 311 42 29	117 147 163 62	155 140 184 29	56 54 69 79	104 96 125 52	16,9																			
0,200 0,348 1,95	319 24 235 67 339 44 30	125 158 175 66	167 151 198 30	59 56 72 86	111 103 134 56	16,4																			
0,220 0,376 1,78	345 25 254 69 366 46 31	134 169 186 70	179 162 213 31	61 59 75 92	119 110 143 59	16,0																			
0,240 0,403 1,63	371 26 273 72 394 47 33	142 179 198 75	191 173 227 33	64 61 78 99	126 116 151 63	15,5																			
0,260 0,428 1,50	397 27 292 75 421 49 34	151 190 210 79	203 184 241 34	67 64 82 105	134 123 160 67	15,0																			
0,280 0,453 1,40	423 28 311 78 448 51 35	160 201 222 83	216 195 256 35	70 67 85 112	141 130 170 71	14,6																			
0,300 0,478 1,30	448 29 330 81 476 53 37	168 212 235 88	228 206 270 37	73 70 89 119	149 138 179 74	14,2																			
0,319 0,500 1,22	473 30 349 84 502 55 38	177 223 246 92	240 216 284 38	76 73 93 125	156 144 188 78	13,8																			
0,340 0,524 1,15	500 31 369 88 530 58 40	186 234 259 97	253 228 300 40	79 76 97 132	165 152 198 82	13,4																			
0,360 0,545 1,09	526 33 388 91 558 60 41	195 246 272 102	265 240 315 41	82 79 101 139	173 159 207 86	13,0																			
0,380 0,567 1,03	552 34 407 95 585 63 43	204 257 285 107	278 251 330 43	86 83 105 146	181 167 217 90	12,6																			
0,400 0,587 0,98	578 36 426 99 613 65 45	214 269 298 112	291 263 345 45	90 86 110 153	189 174 227 95	12,3																			
0,420 0,607 0,93	603 37 445 104 640 68 47	223 281 311 116	304 274 360 47	94 90 115 159	197 182 237 99	11,9																			
0,440 0,626 0,89	629 39 464 108 667 71 49	233 293 325 121	317 286 376 49	98 94 120 166	206 190 247 103	11,6																			
0,460 0,645 0,85	655 40 483 113 695 74 51	243 306 338 127	330 298 391 51	102 98 125 173	215 198 258 108	11,3																			
0,480 0,663 0,81	681 42 502 118 722 78 54	253 318 352 132	343 310 407 54	107 103 131 181	224 206 268 112	10,9																			
0,500 0,681 0,78	707 44 521 124 750 82 56	263 331 366 137	357 322 423 56	112 107 137 188	233 215 279 117	10,6																			
0,530 0,706 0,74	745 47 549 133 791 87 60	279 351 388 146	378 341 448 60	120 115 147 199	247 228 296 124	10,2																			